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# Microwave dielectric properties of LiNb<sub>3</sub>O<sub>8</sub> ceramics with TiO<sub>2</sub> additions

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#### Abstract

The microwave dielectric properties of LiNb<sub>3</sub>O<sub>8</sub> ceramics were investigated as a function of the sintering temperature and the amount of TiO<sub>2</sub> additive. LiNb<sub>3</sub>O<sub>8</sub> ceramics, which were calcined at 750 °C and sintered at 1075 °C for 2 h, showed a dielectric constant ( $\varepsilon_r$ ) of 34, a quality factor ( $Q \times f_0$ ) of 58,000 GHz and a temperature coefficient of resonance frequency ( $\tau_f$ ) of -96 ppm/°C, respectively. The density of the samples influenced the properties of these properties. As the TiO<sub>2</sub> content increased in the LiNb<sub>3</sub>O<sub>8</sub>–TiO<sub>2</sub> system,  $\varepsilon_r$  and  $\tau_f$  of the material were increased due to the mixing effect of TiO<sub>2</sub> phase, which has higher dielectric constant and larger positive  $\tau_f$ . The 0.65LiNb<sub>3</sub>O<sub>8</sub>–0.35TiO<sub>2</sub> ceramics showed a dielectric constant  $\varepsilon_r$  of 46.2, a quality factor ( $Q \times f_0$ ) of 5800 GHz and a temperature coefficient of resonance frequency  $\tau_f$  of near to 0 ppm/°C. © 2005 Elsevier Ltd. All rights reserved.

Keywords: Microwave dielectrics; Dielectric properties; TiO2

## 1. Introduction

The microwave dielectric materials for applications in wireless communication systems such as cellular phones, broadcasting satellites and global positioning systems have been widely studied in the past decade.<sup>1,2</sup> These materials in the range of microwave frequency require a high dielectric constant ( $\varepsilon_r$ ), a high quality factor  $(Q \times f_0)$  and a small temperature coefficient of resonance frequency  $(\tau_f)$ . The recent studies have concentrated on the development of low temperature-cofired ceramics (LTCC) with high conductive internal electrode materials such as silver, copper and their alloys, because of the fabrication of a small resonator within the multilayered integrated circuit.<sup>3,4</sup> Most of the commercial dielectric materials have a high sintering temperature over 1300 °C. To reduce sintering temperature, sintering additives having low-melting points have been generally used in the LTCC systems.<sup>5,6</sup> However, the addition of sintering additives results in an abrupt degrading of the dielectric properties due to the formation of secondary phases. Several dielectric compounds including Nb<sub>2</sub>O<sub>5</sub> and their solid solution have been investigated and the niobate-based materials are tested for microwave dielectrics due to their lower sintering temperature and high quality factors. 7-9

0955-2219/\$ - see front matter © 2005 Elsevier Ltd. All rights reserved. doi:10.1016/j.jeurceramsoc.2005.09.028 The purpose of this work is to examine new niobate dielectric materials combined with lithium oxide that have good microwave dielectric properties and a lower sintering temperature. In order to improve their dielectric constant and temperature coefficient of resonance frequency  $\tau_f$ , the compound is combined with TiO<sub>2</sub> that has a dielectric constant of 104 and a high positive  $\tau_f$  of +450 ppm/°C.<sup>10</sup> The microwave dielectric properties of LiNb<sub>3</sub>O<sub>8</sub>–TiO<sub>2</sub> ceramics have been investigated by varying sintering temperatures and the amount of TiO<sub>2</sub>.

## 2. Experimental procedure

LiNb<sub>3</sub>O<sub>8</sub> compounds were synthesized by the conventional mixed solid oxide method. High purity (99.9%) oxide powders of Li<sub>2</sub>CO<sub>3</sub> and Nb<sub>2</sub>O<sub>5</sub> were used as the starting materials. The powders were weighed and milled with ZrO<sub>2</sub> balls for 12 h in ethanol. The mixed powders were dried and calcined from 650 to 900 °C at rate of 10 °C/min for 2 h, respectively. The calcined powders were mixed with TiO<sub>2</sub> (0.25 to 0.45 mol) in ethanol for 12 h and then dried. These powders were pressed by uniaxial press into pellets of 15 mm diameter and 10 mm thickness under 1000 kg/cm<sup>2</sup> pressure. The pellets were finally sintered from 1025 to 1175 °C at a rate of 10 °C/min for 2 h under air atmosphere.

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The crystalline phases of calcined powders and sintered specimens were analyzed by X-ray powder diffraction (XRD) method (MO3XHF, MAC Science, Japan) for  $2\theta$  in the range  $10^{\circ}$  to  $80^{\circ}$ . The microstructure of the specimens was observed using a scanning electron microscope (LEO420, Cambridge, UK) and the sintered density of the samples was measured by the Archimedes method. The microwave dielectric properties of specimens were measured by the Hakki–Coleman dielectric resonator method with the TE<sub>011</sub> mode.<sup>11</sup> The  $\tau_f$  of the samples was obtained by the cavity method in the temperature range from 25 to  $85 \,^{\circ}C.^{12}$ 

#### 3. Results and discussion

From the XRD analysis of calcined powders showed that a single phase LiNb<sub>3</sub>O<sub>8</sub> compound is formed by heat treatment above 700 °C. However, as the calcination temperature increased, the particle size of the calcined powder increased due to the aggregation of particles. The density of LiNb<sub>3</sub>O<sub>8</sub> ceramics as a function of sintering temperature is shown in Fig. 1. The density increased as the sintering temperature increased up to 1075 °C. The density is decreased slightly above 1100 °C. The density of the specimen sintered at 1075 °C using powder calcined at 75 °C showed the maximum value of 4.84 g/cm<sup>3</sup>.

Fig. 2 shows the SEM micrographs of LiNb<sub>3</sub>O<sub>8</sub> ceramics with various sintering temperatures. The grain size of specimens increases with increasing sintering temperature, but large pores were observed in the specimens sintered above  $1125 \,^{\circ}$ C. The large pores may be related to the volatility of lithium ions during the sintering process. The measurements of weight loss during sintering of the specimens showed no change in weight up to  $1075 \,^{\circ}$ C, but showed a weight loss of about 0.29 wt.% at  $1175 \,^{\circ}$ C. Also, we confirmed that lithium vacancy as a result of its volatility, yields a large abnormal grain growth and changed the grain orientation from the (410) to the (400) plane. From the Fig. 1, as the calcination temperature increased, the apparent density of LiNb<sub>3</sub>O<sub>8</sub> ceramics increased up to 800  $^{\circ}$ C and then decreased at 900  $^{\circ}$ C. Thus, it could be considered that the unre-



Fig. 1. XRD patterns of  $Ca(Li_{1/4}Nb_{3/4})O_3$  calcined specimens with various temperature.



Fig. 2. Dielectric constant and quality factor of  $Ca(Li_{1/4}Nb_{3/4})O_3$  ceramics as a function of sintering temperature.

acted material and aggregation powder prevented the grain size increasing.

XRD patterns of LiNb<sub>3</sub>O<sub>8</sub> ceramics with various sintering temperatures are shown in Fig. 3. The peaks of the (200) and (400) planes were increased with increasing sintering temperature.

Fig. 4 shows microwave dielectric properties of LiNb<sub>3</sub>O<sub>8</sub> ceramics with various calcination temperature as a function of sintering temperature. As the sintering temperatures increased, the dielectric constant increased up to  $1075 \,^{\circ}$ C and then decreased slightly, and the quality factors increased up to



Fig. 3. XRD patterns of  $Ca(Li_{1/4}Nb_{3/4})O_3$  specimens with various sintering temperature.



Fig. 4. Lattice parameters of  $(1 - x)Ca(Li_{1/4}Nb_{3/4})O_3-xCaTiO_3$  system as a function of x mol.

1100 °C and then decreased. These results are very similar to the changes of apparent densities with varying calcination and sintering temperatures. As shown in Fig. 4, the LiNb<sub>3</sub>O<sub>8</sub> ceramics, which were calcined at 750 °C and sintered at 1075 °C for 2 h, showed a  $\varepsilon_r$  of 34, a quality factor ( $Q \times f_0$ ) of 58,000 GHZ and a  $\tau_f$  of -96ppm/°C. In order to improve the dielectric properties of LiNb<sub>3</sub>O<sub>8</sub> ceramics to make a good candidate for microwave dielectrics, it is essential to improve the  $\tau_f$  through the addition of TiO<sub>2</sub> that has a high positive  $\tau_f$  and a high dielectric constant.<sup>11</sup>

X-ray diffraction patterns of (1 - x)LiNb<sub>3</sub>O<sub>8</sub>-*x*TiO<sub>2</sub> ceramics sintered at 1100 °C for 2 h with various TiO<sub>2</sub> concentrations as shown in Fig. 5. XRD patterns could be indexed as two phase mixtures that are composed of the main LiNb<sub>3</sub>O<sub>8</sub> compound and the added TiO<sub>2</sub> compound. With the increase of the TiO<sub>2</sub> concentration in the range of 0.25–0.45 mol, the peak intensity for TiO<sub>2</sub> compound increased steadily. However, the peak positions according to LiNb<sub>3</sub>O<sub>8</sub> were constant due to the absence of solid solutions.

Fig. 5 shows microwave dielectric properties of (1 - x)LiNb<sub>3</sub>O<sub>8</sub>-*x*TiO<sub>2</sub> ceramics sintered at 1100 °C for 2 h as a function of x concentration. As the TiO<sub>2</sub> concentration increase from 0.25 to 0.45 mol, the dielectric constant and temperature coefficient of resonance frequency  $\tau_f$  increased from 40.5 to 55.8 and from -52 to 43 ppm/°C, respectively. This can be explained by the logarithmic mixing rule of properties in mixture ceramics between LiNb<sub>3</sub>O<sub>8</sub> having a dielectric constant of 34, a  $\tau_f$  of  $-96 \text{ ppm/}^{\circ}\text{C}$  and by TiO<sub>2</sub> having a dielectric constant of 104 and a  $\tau_f$  of +450 ppm/°C. These results are well agreed with the results related to rutile and its compounds by Haga K. et al.<sup>10</sup> However, the quality factor  $(Q \times f_0)$  values decrease with an increasing TiO2 concentration. This could be considered due to the fact that the phase boundary in mixture ceramics was affected by an increase of interface loss resulting from the generation of anharmonic vibration that decreases the quality factor.<sup>13</sup>



Fig. 5. Dielectric constant, quality factor and TCF of  $(1 - x)Ca(Li_{1/4}Nb_{3/4})O_3 - xCaTiO_3$  system as a function of x mol.

In conclusion, 0.65LiNb<sub>3</sub>O<sub>8</sub>–0.35TiO<sub>2</sub> ceramics sintered at 110 °C have good microwave dielectric properties with a dielectric constant of 46.2, quality factor ( $Q \times f_0$ ) values of 5800 GHz and a  $\tau_f$  of near to 0 ppm/°C.

# 4. Conclusion

LiNb<sub>3</sub>O<sub>8</sub> ceramics, which were calcined at 750 °C and sintered at 1075 °C for 2 h, showed a  $\varepsilon_r$  of 34, a  $Q \times f_0$  of 58,000 GHz and a  $\tau_f$  of -96 ppm/°C. These properties were influenced by changes in sintered densities due to the calcination temperature and lithium evaporation.

As the TiO<sub>2</sub> content increases in LiNb<sub>3</sub>O<sub>8</sub>–TiO<sub>2</sub> systems, the dielectric constant and  $\tau_f$  increased due to the mixing effect of the TiO<sub>2</sub> phase. The  $\varepsilon_r$  of 46.2, a quality factor of 5800 GHz and a  $\tau_f$  of near 0 ppm/°C were obtained for the 0.65LiNb<sub>3</sub>O<sub>8</sub>–0.35TiO<sub>2</sub> ceramics sintered at 1100 °C.

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